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Bandgaps of bent and buckled carbon nanotubes

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Carbon nanotube's large electro-mechanical coupling and robustness makes them attractive for applications where bending and buckling is present. But the nature of this coupling is not well understood. Existing theory treats only weak and homogeneous deformations. We generalize it and derive close-form expressions for bandgaps under non-homogeneous deformation. The theory is first compared with a number of published DFT simulations, and then applied to the specific case of bending and buckling – where a kink is present. In the pre-buckling regime, bandgaps change $\propto \pm \kappa^4$ where κ is the bending curvature; inside the kink at post-buckling, while near criticality, the kink is shallow and the gap is $\propto \pm \kappa^{1/2}$, where the sign depends on the chiral integers. For a deeper kink but still with an open cross-section, both the bandgap and local Fermi energy strongly downshift.

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I. INTRODUCTION

Three decades after its discovery, carbon nanotubes has yet to realize its potential. A slow but continuous progress in fabrication may, however, make feasible a growing number of application ideas suggested early-on but never pursued due to economical and fabrication considerations. One such application class would be electromechanics – as it relies on the coupling between two of the tube's unique features: one dimensional conductivity and elasticity. This coupling is most extreme in the kink of a buckled tube. The kink's size is of the order of the tube diameter, but what is its electronic structure and how will it affect transport as a function of further bending? Currently, the answer is far from being clear. The kink is a point of large curvature and non-homogeneous deformation, and, as we show below, non of these is theoretically well understood.

The elementary theory predicting bandgaps in carbon nanotubes is based on zone-folding of graphene [5]. It classifies tubes as metallic or semiconducting – with bandgaps $\propto 1/R$ according to their chiral vectors (appendix B). Corrections $\propto 1/R^2$ due to circumferential curvature [6][7], predict that metallicity is lifted in all but one type: the (achiral) armchair tubes. An obvious prediction is the bandgaps would trend higher at ever smaller radii. But DFT computations (table II) clearly show that the opposite is true: the narrowest tubes are in-fact gapless; it is qualitatively attributed to a curvature-induced hybridization of the $\sigma - \pi$ bands, but it has not been quantitatively incorporated into existing theory.

This should also play a critical role in the kink, where very large curvature is present. Thus, our first aim is to understand the effects of large curvature quantitatively. But the other characteristic of the kink – its large and non-homogeneous deformation, needs also to be accounted for.

The modulation of bandgaps due to homogeneous deformation is well-known [8][9]. But that is not sufficient, even for weak bending without kinks; the strain profile in bending is asymmetric about the neutral line, i.e: pure bending has no *net* strain; this would imply, according to existing theory, no change in the bandgap – a clear contradiction with simulations [10][11].

Thus, in order to reconcile theory and simulations and derive closed-form solutions for the bandgap in the kink, we need, in addition to incorporating effects of large curvature, extend the theory of homogeneous deformation into the non-homogeneous domain. For completeness, however, we start by recapping the existing, homogeneous, theory of bandgaps in carbon nanotubes.

Symbol	Definition
ε	axial strain
ζ	axial torsion
$\kappa_{ heta}$	circumferential curvature
κ	bending curvature
κ_z	local axial curvature
$\kappa^{ m cr}$	bending curvature at buckling point
$b=3.5~{\rm eV/\AA}$	$d\gamma/dl$, where l is bond length
$\gamma = 2.7 \mathrm{eV}$	π -orbitals overlap energy
n,m	nanotube's chiral integers; $m \leq n$
p,q	integers; $n - m = 3q + p$ (B1)
v,w	in-plain and out-of-plain deformation vectors, respectively (appendix A)
c_h	ac_h is the circumference; $c_h = \sqrt{n^2 + nm + m^2}$
R	nanotube's radius, $2\pi R = ac_h$
$\nu = 0.19$	Poisson ratio [1]; $\nu = 0.34$ according to [2]; see also [3], [4]
$t = 0.66 \text{\AA}$	effective thickness $[1]$; 0.75Å according to $[2]$; see also $[3]$, $[4]$
α	chiral angle; $\cos 3\alpha = (2n^3 - 2m^3 + 3n^2m - 3nm^2)/2c_h^3$
ξ	ovalization parameter (eq. A3)
$\xi^{ m cr}$	critical ovalization $-2/9$
E_g^{π}	bandgap due to the π -band alone
$E_g^{\rm pre}$	bandgap at the pre-buckling regime
E_g^{post}	bandgap at the post-buckling regime
E_s	energy of the singlet-band above graphene's Fermi level
C_s	coefficient for E_s (appendix C); $C_s = 8 \sim 12 \text{eV} \text{ Å}^2$
$R_{10,0}$	the radius of the (10,0) tube; $R_{10,0} = 4\text{\AA}$
R_c	upper critical radius; above it, unhybridized $\pi\text{-band}$ structure is valid
R_v	lower critical radius; below it, bandgaps are zero
\vec{K}_F	Fermi points
δk_y	lateral shift of the Fermi points
\vec{l}, l_y, l_z	nearest neighbor bond vectors and their lateral and axial projections
U	elastic energy per unit length

Table I: List of symbols and values.

II. HOMOGENEOUS DEFORMATION

In this section we start by recasting the tight-binding π -band theory of homogeneous deformation [7][8][9] in a formalism that will be used here for non-homogeneous fields; we then consider deviations from the pure π -band, which become dominant at very small radii, or large circumferential curvatures.

A. General formulation

The fundamental theory of bandgaps in carbon nanotubes, reproduced here in appendix B, gives increasingly erroneous gaps as the nanotubes' diameter decreases. The reason lies in a symmetry of graphene that is broken in nanotubes: isotropy of bonds. The assumption that the three nearest overlap integrals are equal (B5), does not fully hold in nanotubes, since neighboring π -orbitals are not parallel, nor are neighboring atoms equally separated. Discarding thus assumption (B5), we find the new Fermi points by solving eq. (B3) while allowing small changes to the overlap integrals $\delta \gamma_j$. The solution yields [9],

$$\delta k_y = \frac{\operatorname{sgn}[1-2p]}{2\pi R\sqrt{3}\gamma} \times (1)$$
$$\left((m-n)\delta\gamma_1 + (2n+m)\delta\gamma_2 - (n+2m)\delta\gamma_3 \right).$$

where δk_y is the shift of the Fermi points along the circumferential coordinate y, relative to their zone-folding position (B6). Now, knowing the $\delta \gamma_j$ in (1), translates, through the linear spectrum (B7) to the following gap equation,

$$E_g^{\pi} = \frac{a\gamma}{R} \left| \frac{p}{\sqrt{3}} + \sqrt{3}R\delta k_y \right|.$$
 (2)

The superscript π serves to label it as a pure π -band, in contrast to the hybridized band to be considered later.

B. Small circumferential curvature

One of the symmetries present in graphene but not in nanotubes is the parallel nature of the π -orbitals. In nanotubes, neighboring π -orbitals are misaligned by a small angle: if \vec{l}_j is the vector connecting them (eqs. B4), and l_{jy} is its y-component, the angle is $\beta_j = l_{jy}/R$. Then $\gamma_j \rightarrow \gamma \cos^2 \beta_j$ – which gives

$$\delta\gamma_j = -\gamma \frac{l_{jy}^2}{2R^2}.$$
(3)

Now plugging this in eqs. (1) and (2) yields

$$\frac{E_g^{\pi}}{\gamma} = |p| \frac{a}{R\sqrt{3}} + \text{sgn}[1-2p] \frac{a^2 \cos 3\alpha}{16R^2}, \qquad (4)$$

where $\cos 3\alpha$ is given in table I. E_g^{π} includes two effects over the π -band: the 1/R-rule of the fundamental gap, and the $1/R^2$ correction due to curvature. Here, in contrast with the fundamental prediction, "metallic" tubes (p = 0) do open small gaps, and their $1/R^2$ trend, as given by eq. (4), was experimentally verified on a few large zigzag tubes [17].

C. Large circumferential curvature

Predictions from the gap equation (4) can be compared with MD-simulations: table (II) lists results from simulations run by different groups, and fig. (1) compares them with eq. 4 for zigzag tubes. Two opposite trends in fig. 1 are evident: in tubes larger then (10,0), the gaps seen to agree with eq. (4), but smaller tubes reverse this trend and quickly reach metallicity.

The reason for this was pointed out early-on by Blase et al. [18]; it concerns a singly-degenerate band (termed below: S-band) which is a $\pi^* - \sigma^*$ hybridization. In graphene, the S-band is positioned very high above Fermi level; but since hybridization is $\propto 1/R^2$, the S-band downshifts with decreasing radius, and at a certain radius it crosses the conduction band. This radius was found [13] by DFT to be the radius of the (10,0) tube. i.e. in this tube, the S and conduction bands overlap. Its radius is,

$$R_{10,0} \approx 4\text{\AA}.$$
 (5)

At smaller radii, the S-band cuts through the gap (see for example: (8,0) and (7,0) in fig. 1) until, at the smallest radii, it vanishes entirely (see for example: (6,0), (5,0)and (4,0) in table II).

Let E_s be the energy, relative to Fermi level, of the S-band at the K-points. Its $1/R^2$ dependence was found by DFT computations to scale by a factor of $C_s \sim 8$ (ev·Å²) [15] (see also appendix C). But the hybridization near the K-points scales also with chirality $\propto \cos 3\alpha$ [19]. Then, the functional form of E_s is

$$E_s = \frac{1}{2} E_g^{\pi}(10,0) - C_s \left(\frac{1}{R^2} - \frac{1}{R_{10,0}^2}\right) \cos 3\alpha, \quad (6)$$

where $E_g^{\pi}(10,0) \approx 0.93 \text{eV}$ – is the un-hybridized bandgap of the (10,0) tube. The radii, R_c and R_v , at which the *S*-band cross the conduction and valence bands, respectively, can be found by setting

$$E_s - \frac{1}{2} E_g^{\pi} = 0, \quad R = R_c, \quad (7)$$
$$E_g^{\pi} - C_s \left(\frac{1}{R^2} - \frac{1}{R_{10,0}^2}\right) \cos 3\alpha = 0, \quad R = R_v. \quad (8)$$

This yields

$$R_{c} = \frac{1}{2a_{c}} \left(b_{c} + \sqrt{b_{c}^{2} + 4a_{c}c_{c}} \right), \text{ where }$$
(9)

$$a_{c} = \frac{C_{s}\cos 3\alpha}{R_{10,0}^{2}} + \frac{E_{g}^{\pi}(10,0)}{2},$$

$$b_{c} = |p|\frac{\gamma a}{2\sqrt{3}},$$

$$c_{c} = \left(C_{s} + \text{sgn}[1 - 2p]\frac{\gamma a^{2}}{32} \right)\cos 3\alpha,$$

and,

$$R_v = \frac{1}{2a_v} \left(-b_v + \sqrt{b_v^2 + 4a_v c_v} \right), \text{ where } (10)$$

$$a_v = \frac{C_s \cos 3\alpha}{R_{10,0}^2},$$

$$b_v = |p| \frac{\gamma a}{\sqrt{3}},$$

$$c_v = \left(C_s - \operatorname{sgn}[1 - 2p] \frac{\gamma a^2}{16} \right) \cos 3\alpha,$$

Eqs. 9, 10 yield three pairs of R_c, R_v – a pair for each p. These radii thus delimit the bandgap regimes: large tubes $(R > R_c)$ have their bandgap expressed by the usual π -band, E_g^{π} (eq. 4); smaller tubes $(R_v < R < R_c)$ have the S-band between the valence and the conduction bands, hence the bandgaps in this regime become

$$E_g^{\pi} - C_s \left(\frac{1}{R^2} - \frac{1}{R_c^2}\right) \cos 3\alpha \tag{11}$$

where E_g^{π} is the non-hybridized (π -band) gap (4) and $C_S \sim 8 \text{ eV} \cdot \text{Å}^2$ is a semi-empirical scaling factor (see appendix C). The smallest tubes ($R \leq R_v$), on the other

n,m	ref. [12]	ref. [13]	ref. [14]	ref. [15]	ref. [16]	eqs. (12)
4, 0	0	0	0	0	0	0
5,0	0	0	0	0	0	0
5, 1	-	-	-	0.05	0.13	0.06
5,3	-	-	-	1.2	1.18	1.12
5,4	-	-	-	1.12	-	1.14
6, 1	-	-	-	0.43	0.41	0.71
6, 2	-	-	-	0.7	0.67	0.74
6, 4	-	-	-	1.1	1.09	1.07
7,0	0.66	0.34	0.19	0.2	0.21	0.474
8,0	1.05	0.487	0.73	0.6	0.59	0.853
10, 0	0.96	0.87	0.88	0.8	0.77	0.913
11, 0	1.1	0.8	1.13	-	0.93	0.945
13,0	0.79	-	0.73	-	0.64	0.714
14, 0	0.85	-	0.9	-	0.72	0.733
16, 0	-	0.57	0.61	-	0.54	0.586
17,0	-	-	-	-	0.58	0.599
19,0	-	-	-	-	0.46	0.497
20, 0	-	-	-	-	0.5	0.506

Table II: Bandgaps (in eV) of semiconducting tubes.

hand, are metallic since the S-band crossed the valence band. The bandgaps are then given by,

$$E_{g} = \begin{cases} \frac{|p|\gamma a}{R\sqrt{3}} + \mathrm{sgn}[1-2p] \frac{\gamma a^{2}}{16R^{2}} \cos 3\alpha, & R \ge R_{c}, \\ \frac{|p|\gamma a}{R\sqrt{3}} + \left(\mathrm{sgn}[1-2p] \frac{\gamma a^{2}}{16} - C_{s}\right) \frac{\cos 3\alpha}{R^{2}} + \\ + \frac{C_{s} \cos 3\alpha}{R_{c}^{2}}, & R_{v} < R < R_{c}, \\ 0, & R \le R_{v}. \end{cases}$$
(12)

This is the complete solution for bandgaps under no deformation. It gives comparable results with various published simulations (table II); fig. (1) plots the bandgaps of zigzag tubes according to eqs. (12) and the simulations.

D. Strain

Having found the general gap equations (12), we wish to find its response to homogeneous deformation, such as axial strain, ϵ_z . Following the procedure in [9] and [8], the bond vectors (B4) $\vec{l}_j = (l_{jy}, l_{jz})$, transform as

$$\vec{l}_j \to \mathbf{D}\vec{l}_j,$$
(13)

where \mathbf{D} is a deformation matrix, given, for axial strain, by

$$\mathbf{D} = \begin{pmatrix} 1 - \nu \epsilon_z & 0\\ 0 & 1 + \epsilon_z \end{pmatrix} \tag{14}$$

where ν is the tube's Poisson ratio (table I). This deformation alters the bond distances, $|l_j| \rightarrow |l_j| + \delta |l_j|$, which in turn, alters the overlap integrals $\gamma_j \rightarrow \gamma_j + \delta \gamma_j \approx \gamma_j + b\delta |l_j|$, where $b \approx 3.5 \text{eV/Å}([9])$. This yields, using (13) and (14),

$$\delta\gamma_j = \frac{\epsilon_z b\sqrt{3}}{a} (-\nu l_{jy}^2 + l_{jz}^2). \tag{15}$$

Now plugging the above three $\delta \gamma_j s$ in eq. (1), we get

$$\delta k_y = \frac{1}{2\gamma} \operatorname{sgn}[2p-1]\epsilon_z b(1+\nu) \cos 3\alpha.$$
(16)

This equation is the explicit form of eq. (1) under axial strain. On substitution in eq. (2) we get,

$$E_g^{\pi} = \frac{|p|\gamma a}{R\sqrt{3}} + \left(\frac{\gamma a^2}{8R^2} - \epsilon_z(1+\nu)ba\sqrt{3}\right) \\ \times \frac{1}{2}\mathrm{sgn}[1-2p]\cos 3\alpha \tag{17}$$

But as discussed above, at small radii the π -band is crossed by and hybridized by the singlet band, downshifting the gap (11), until it vanishes. This yields the three regimes, as in eqs. 12,

$$E_{g}(\epsilon_{z}) = \begin{cases} E_{g}^{\pi}, & R \ge R_{c}, \\ E_{g}^{\pi} - C_{s} \left(\frac{1}{R^{2}} - \frac{1}{R_{c}^{2}}\right) \cos 3\alpha, & R_{v} < R < R_{c} \\ 0, & R \le R_{v}. \end{cases}$$
(18)



Figure 1: Bandgaps of semiconducting zigzag tubes vs. $1/R^2$. The two type of semiconductors, $p = \pm 1$ (for definition see eq. B1), are marked with red and blue colors, respectively. The integers above/below the x-axis are aliases for the zigzags (n, 0); they are positioned under their DFT bandgaps (dots connected by vertical lines); the dots, labels a, b, c, d and e, refer to the correspond publications (see table II). The full curves are plots of eq. (12). The broken curve is the zone-folding "1/R-rule" (eq. B8). It is evident that, for large tubes, this rule holds with slight modification, but for smaller tubes the gap reverses trend and finally vanishes. The critical radius of trend reversal is ~ 4.2Å, which lies between the tubes (11,0) and (10,0). The regions A, B and C correspond to the regimes $R \ge R_c, R_v < R < R_c$ and $R \le R_v$, respectively.

An unequivocal experimental verification of eqs. (18) is not known to us, but the strain and chiral angle dependence of E_g^{π} was probed by [20] and qualitatively confirmed. Simulations of compressive strain in zigzag tubes [21] appear to agree with both the linear behaviour and *p*-dependence of E_q^{π} as given by eq. (17).

E. Torsion

To find the bandgap under an axial torsion, ζ , we start by setting its deformation matrix,

$$\mathbf{D} = \begin{pmatrix} 1 & 0 \\ -\zeta & 1 \end{pmatrix}. \tag{19}$$

following the steps applied to strain (14-18), we get for the un-hybridized π -bandgap

$$E_g^{\pi} = \frac{|p|\gamma a}{R\sqrt{3}} + \frac{1}{2}\mathrm{sgn}[1-2p] \times \left(\frac{\gamma a^2 \cos 3\alpha}{8R^2} - \zeta ba\sqrt{3} \sin 3\alpha\right).$$
(20)

As with strain, this bandgap is valid only in the regime where the S-band is above the (π) conduction band $(R > R_c, \text{ eq. 9})$. Otherwise, hybridization with the S-band, as discussed above, strongly downshifts it. In complete equivalence to strain (18), the torsion bandgaps yield the general formulae,

$$E_{g}(\zeta) = \begin{cases} E_{g}^{\pi}, & R \ge R_{c}, \\ E_{g}^{\pi} - C_{s} \left(\frac{1}{R^{2}} - \frac{1}{R_{c}^{2}}\right) \cos 3\alpha, & R_{v} < R < R_{c}, \\ 0, & R \le R_{v}, \end{cases}$$
(21)

where E_g^{π} is the torsion-modified bandgap (20) of the π -band alone.



Figure 2: R_c and R_v (eqs. 9, 10) for p = +1 (red) and p = -1 (blue) tubes. The bandgap of tubes whose radius lies above R_c is not affected by hybridization. While the gap of tubes that lie between R_c and R_v (colored background) is affected – and reduced. The couple of tubes lying below R_v have zero gap. The regions A, B and C, as in fig. (2), correspond to the regimes $R \ge R_c$, $R_v < R < R_c$ and $R \le R_v$, respectively.

III. NON-HOMOGENEOUS DEFORMATION

So far, the bandgaps found here concerned tubes with circular cross-section and axial uniformity. But that is not the case in many situations. The cross-section of a bent tube, for example, is compressed in the inner side of bending-curvature and stretched in the other. Its deformation profile is thus varying throughout the unit cell of the tube. To treat this and similar cases, we need to generalize the formalism of section (II A) to include these non-homogeneous deformation.

A. General formulation

In the theoretical treatment of section II A, it was tacitly assumed that the deformation is everywhere identical. This allowed us to deform a single graphene unit cell and extract the shift of the Fermi points from there. But that can not be done when the deformation varies throughout the tube's unit cell. Hence, if N is the number of graphene unit cells in the nanotube unit cell, the position of the Fermi points, normally found by solving eq. (B3), is now found by the sum over the N A-atoms

$$\sum_{A=1}^{N} \sum_{j=1}^{3} \gamma_{Aj} e^{i\vec{k}\cdot\vec{l}_{Aj}} = 0.$$
 (22)

As before, we seek the lateral shift ΔK_y which is a simple sum of $\delta k_u(A)$ (1) of the constituent A-atoms,

$$\Delta K_y = \frac{1}{N} \sum_{A=1}^{N} \delta k_y(A) \tag{23}$$

where $\delta k_y(A)$ is now a function of the local strain and curvature at the position of the corresponding A'th atom.

The overlap integrals γ_{Aj} in eq. 22 are functions of the in-plain and out-of-plain deformations of the bond vectors \vec{l}_{Aj} . In-plain deformation, such as strain and torsion, changes the γ 's by changing the distance between neighboring π -orbitals, while out-of-plain deformation, such as the tube's curvature, lowers the γ 's by having the orbitals misaligned.

In this section we include both deformations without assuming their homogeneity throughout the tube. First consider curvature. On a plain cutting through the cross section, let $\pm \phi_y$ be the angles between neighboring π orbitals and the normal passing through the middle point between them; and let $\pm \phi_z$ be the corresponding angles projected on a plain along the axis. If $\kappa_{\theta}, \kappa_z$ are the coordinate curvatures defined as (radius of curvature)⁻¹ within the respective plains, and writing bond vectors (B4) in the form $\vec{l} = l_y \hat{y} + l_z \hat{z}$, one gets $|\phi_y| = |l_y|\kappa_{\theta}/2$ and $|\phi_z| = |l_z|\kappa_z/2$. The overlap integral becomes $\gamma \rightarrow \gamma \cos \phi_y \cos \phi_z$, so that for small change $\delta\gamma$ is given by

$$\delta\gamma^{\rm curv} = -\frac{\gamma}{8} \left(l_y^2 \kappa_\theta^2 + l_z^2 \kappa_z^2 \right). \tag{24}$$

Let us now include local axial strain. Its deformation matrix, given by eq. (14), induces a change $\delta \gamma^{\text{strain}}$ according to eq. (15). The total change in the local overlap integrals is then,

$$\delta\gamma = \delta\gamma^{\rm curv} + \delta\gamma^{\rm strain} \equiv l_y^2 D_y + l_z^2 D_z, \qquad (25)$$

where

$$D_y = \frac{\nu b\sqrt{3}}{a} \epsilon_z - \frac{\gamma}{8} \kappa_\theta^2, \qquad (26)$$

$$D_z = -\frac{b\sqrt{3}}{a}\epsilon_z - \frac{\gamma}{8}\kappa_z^2.$$
 (27)

Substituting l_y, l_z from eqs. B4 in eq. (25) we find,

$$\delta\gamma_{A1} = \frac{a^2}{4c_h^2} \left((n+m)^2 D_y + \frac{(n-m)^2}{3} D_z \right)$$

$$\delta\gamma_{A2} = \frac{a^2}{4c_h^2} \left(m^2 D_y + \frac{(2n+m)^2}{3} D_z \right)$$

$$\delta\gamma_{A3} = \frac{a^2}{4c_h^2} \left(n^2 D_y + \frac{(2m+n)^2}{3} D_z \right)$$
(28)

Inserting now eqs. 28 in 1

$$\delta k_y = \operatorname{sgn}[1-2p] \frac{a(D_z - D_y)}{2\sqrt{3\gamma}} \cos 3\alpha.$$
 (29)

This, by eq. (23), we sum over all A-atoms in the tube's unit cell,

$$\Delta K_y = \text{sgn}[1 - 2p] \frac{a \cos 3\alpha}{2\sqrt{3}\gamma N} \sum_{A=1}^{N} (D_z - D_y).$$
(30)

Converting now the sum to an integral,

$$\Delta K_y = \operatorname{sgn}[1-2p] \frac{a\cos 3\alpha}{2\sqrt{3}\pi\gamma} \int_0^{2\pi} (D_z - D_y) d\theta. \quad (31)$$

Once Δk_y is known, the energy gap follows immediately from the dispersion relation (B7) and the zone-folding gap (B8),

$$E_g^{\pi} = \frac{a\gamma}{R} \left| \frac{p}{\sqrt{3}} + \sqrt{3}R\Delta k_y \right|.$$
(32)

Equation (32) is the non-homogeneous version of eq. (2), and likewise, it includes only the π -band. But now

the non-homogeneity of the deformation must be included through the integration in (31), we thus replace $1/R^2$ in eq. (6) accordingly,

$$\frac{1}{R^2} \to \frac{1}{2\pi} \int_0^{2\pi} \kappa_\theta^2 \, d\theta. \tag{33}$$

Now the full set of gap equations for non-homogeneous deformation can be obtained by using equations (12) with the substitution of eq. (32) for E_g^{π} , and a replacement as in (33).

B. Elastic theory of bending and buckling

The structural properties of bending and buckling of SWCNT's had been simulated [1][4][22][23][24] [25] and experimented [22][2] in the years following nanotubes' discovery. They established that SWCNT's abide by continuous elasticity theory given an effective wall "thickness" of 0.66Å (values by other groups in table I). Which is always much smaller then the tube's diameter. Hence SWCNTs are also "shells"; precisely: *slender cylindrical shells*.

The elastic theory of their bending, up to the onset of buckling, was developed a century ago by Brazier [26]. It is summarized in appendix A for convenience.

Brazier's fundamental insight was that, in order to reduce shear under bending, the tube's circular crosssection is ovalized (fig. 5). At the critical bending, however, the elastic cost of increased ovalization exceeds the reduction in shear energy – and the tube buckles.

In the pre-buckling regime, Brazier's theory gives the exact shape of the ovalized cross-section, parametrized by ξ in fig. 5. It also predicts, as reproduced in the appendix, that at the onset of buckling the ovalization is $\xi = 2/9$ (eq. A19), independent of Poisson's ratio or other material properties.

At the post-buckling regime, the elastic energy associated with ovalization (beyond the buckling point) is

$$\Delta U = G(\xi - \xi^{\rm cr})^2, \qquad \xi \ge \xi^{\rm cr} \tag{34}$$

where G is a constant. But MD simulations [1][22] showed that in this regime, the post-buckling energy density is linear with bending curvature, that is

$$\Delta U = Q(\kappa - \kappa^{\rm cr}), \qquad \kappa \ge \kappa^{\rm cr}, \tag{35}$$

where Q is a constant and κ^{cr} is the critical bending curvature (A17). Comparing eqs. (34) and (35),

$$\xi - \xi^{\rm cr} = \frac{Q}{G} (\kappa - \kappa^{\rm cr})^{1/2}, \qquad \kappa \ge \kappa^{\rm cr}.$$
(36)

This gives the proportionality relation in the kink, between bending curvature, κ , and ovalization of the crosssection, ξ . Outside of the kink, however, This implies that outside the kink, strain is independent of further bending – its energy is absorbed in the kink, which acts as a hinge between the two fixed sections of the tube [27][28].

A realistic closed-form model of such kinks is not known to us. But simulations revealed [29] that for bending not too far from the onset of buckling, where the kink is shallow, the cross-section is shaped as an oval, just as the oval parametrized by Brazier in the pre-buckling regime (appendix A). Since this regime is an intermidiate stage between the onset of buckling and a fully developed kink, it is also called the *transient-regime* [29].

This regime ends when the opposite walls of the kink approach contact; at this point the distance between the walls is comparable with the inter-planar distance in graphite $d_g \sim 3.35$ Å. In Brazier's parametrization this distance is $2R(1-\xi)$ (fig. 5). We thus have

$$\xi^{\text{close}} = 1 - \frac{d_g}{2R},\tag{37}$$

where the superscript *close* signifies the closing of the cross-section in the kink and bringing the opposite walls to near contact. ξ^{close} marks the upper limit of flattening within which our analysis is expected to hold. For a tube of 1nm in diameter, $\xi^{\text{close}} \sim 2/3$.

The cross-section of a kink in the transient regime is thus bounded by the onset of buckling, at $\xi = 2/9$, and ξ^{close} , having the curvature dependence according to eq. (36), namely,

$$\xi^{\text{kink}} = \frac{2}{9} + \left(\frac{\kappa - \kappa^{\text{cr}}}{\kappa^{\text{close}} - \kappa^{\text{cr}}}\right)^{1/2} \left(\xi^{\text{close}} - \frac{2}{9}\right), \quad (38)$$

where $\kappa^{\text{cr}} \le \kappa \le \kappa^{\text{close}}.$

Equation 38 demonstrates the evolution of ξ^{kink} , being continuous at the onset of buckling $\xi^{\text{kink}} = 2/9$, through further bending where $\xi^{\text{kink}} \propto \kappa^{1/2}$, and finally reaching the closing point at ξ^{close} .

Having found ξ^{kink} , the circumferential curvature in the center of the kink is given, as in the pre-buckling regime, through the out-of-plane deformation w (eq. A3) and eq. (A4) by,

$$\kappa_{\theta}^{\text{kink}} = \frac{1}{R} (1 + 3\xi^{\text{kink}} \cos 2\theta). \tag{39}$$

The axial strain under bending, ϵ_z , is anti-symmetric with respect to the neutral plane; it thus has no contribution to the integration in eq. (31); also the bending curvature, $\kappa_z \ll 1/R$, can be neglected in the integration (31), although, as will be shown next, it affects κ_{θ} . Hence we set in eqs. (26-27) $D_y = -\frac{\gamma}{8}\kappa_{\theta}^2$ and $D_z = 0$.

According to the elastic theory of bending (appendix A), the circular cross-section of a bent tube becomes increasingly oval with bending. Parametrizing the ovalization by ξ (fig. 5), let the circumferential integral of the square of the curvature (eq. A4) be I, then

$$I \equiv \frac{1}{2\pi} \int_0^{2\pi} \kappa_\theta^2 d\theta = \frac{1 + \frac{9}{2}\xi^2}{R^2},$$
 (40)

where the pre-buckling regime corresponds to $0 \le \xi \le 2/9$. The point $\xi = 2/9$ corresponds to the critical curvature at buckling (eq. A19).

C. Bending – pre-buckling

Since the ovalization in this regime is a quadratic function of the bending curvature (eq. A14), the replacement rule (33) becomes,

$$I = \frac{1}{R^2} \to \frac{1 + \frac{9}{2}\xi^2}{R^2} = \frac{1}{R^2} \left(1 + (L\kappa)^4\right).$$
(41)

where

$$L \equiv (1 - \nu^2)^{1/2} \left(\frac{9}{2}\right)^{1/4} \frac{R^2}{t}$$

where ν and t are given in table I. With this replacement, eqs. (12) yield the explicit gap equations under bending,

$$E_g^{\rm pre}(A) = \frac{|p|a\gamma}{R\sqrt{3}} + \frac{\operatorname{sgn}[1-2p]\,\gamma a^2}{16R^2} \left(1 + (L\kappa)^4\right) \cos 3\alpha,\tag{42}$$

where the superscript "pre" refers to pre-buckling and "A" corresponds to $R \ge R_c$. Region B then corresponds to radii in the range $R_v < R < R_c$; the bandgaps there are,

$$E_g^{\rm pre}(B) = \frac{|p|a\gamma}{R\sqrt{3}} + \frac{(\text{sgn}[1-2p]\,\gamma a^2 - 16C_s)}{16R^2} \times (1 + (L\kappa)^4)\cos 3\alpha, \qquad (43)$$

and finally, at the smallest radii range, C, where $R \leq R_v$,

$$E_g^{\rm pre}(C) = 0. \tag{44}$$

Eqs. (42–44) give the bandgaps for all radii in the pre-buckling regime; the three ranges of radii, A, B and C are also shown in figs. (1) and (2). These equations reveal that, depending on the sign of p, bending may increase, decrease or even close the gap – a prediction which also coincides with simulations of ovalized crosssections [30][31].

D. Bending – critical curvature

At the critical point of buckling the ovalization parameter $\xi^{\rm cr} = 2/9$ (appendix A). This gives for the rhs of eq. (40) $I = \frac{11}{9R^2}$. It is interesting to note that for metallic tubes (p = 0), replacing the rhs of eq. 41 for this value of I and substituting in eq. 42, gives,

$$\frac{\Delta E_g(\kappa = \kappa^{\rm cr})}{E_g(\kappa = 0)} = \frac{2}{9}, \qquad p = 0, \ R \ge R_c. \tag{45}$$

This universal ratio for metallic tubes (armchair tubes excluded as their $E_g = 0$), relates their bandgaps at the critical point of buckling with their straight value. Figure (3) depicts this in terms of the change in the bandgap as a function of the bending curvature.



Figure 3: Bandgap vs. bending curvature for primary metallic tubes, p = 0, (armchairs excluded). E_g is the gap of the straight tube (eq. 4); ΔE_g is the additional gap due to pure bending (42-44) and (48-50); κ is the bending curvature and κ^{cr} is the critical curvature (eq. A17). In the pre-buckling regime the gap $\propto \kappa^4$ up to

buckling point where it is higher by a factor of 2/9

compared with the straight state (eq. 45). At post-buckling, the bulk relaxes by a small amount δ (46) and remains there, while in the kink it is $\propto (\kappa - \kappa^{\rm cr})^{1/2}$.

E. Bending – post-buckling

Since the elastic bending moment reaches maximum at buckling point [32], the buckling transition is accompanied by a small relaxation of the strain energy [1] [22] outside of the kink. This relaxation also lowers Brazier's ovalization and, as will be shown next, also the associated bandgap.

The pre-buckling elastic energy per unit length (eq. A16) can be approximated near critical point by $U \sim \frac{1}{2}R^3 t E \pi \kappa^2$. Denoting the post-buckling elastic energy relaxation per unit length by ΔU , the relaxed state corresponds to a lower curvature given by $\delta \kappa = \Delta U/(\partial U/\partial \kappa)$. The associated shift in the band-gap ratio $\Delta E_g/E_g$ for metallic tubes (eq. 45) can be found by expanding this ratio near on the pre-buckling side of criticality, where the function is well behaved $\delta(\Delta E_g/E_g) = \partial(\Delta E_g/E_g)/\partial \kappa \delta \kappa$ where the derivative is taken at the buckling curvature $\kappa = \kappa^{\rm cr}$ (eq. A17); this yields,

$$\delta\left(\frac{\Delta E_g}{E_g}\right) = \frac{R\Delta U}{3\pi D},\tag{46}$$

where D is the elastic rigidity (eq. A11) and ΔU is the elastic energy relaxation per unit length at buckling. Equation 46 links, for metallic tubes, the mechanical post-buckling relaxation step with the post-buckling electronic band-gap change in the bulk. In contrast with the relaxation in the bulk, the bandgap in the kink is now a stronger function of bending curvature κ ; this can be seen by applying the replacement rule (33) on the gap eqs. (12) using $\kappa_{\theta}^{\text{kink}}$ (39).

Staying, as we do throughout this work, within the transient regime where the kink remains open ($\xi^{\text{kink}} <$ ξ^{close} , eqs. 37, 38), bandgaps in the kink fall in three regimes; these are related to the three regimes in straight tubes (fig. 1), corresponding to whether the gap is determined by the π -band alone (regime A), the modified regime where the singlet (S) band hybridizes with the π band (regime B), or zero – where the S-band downshifted enough to completely close the gap (regime C). All tubes fall into one of these regimes depending on their diameter (largest to smallest – regimes A to C, respectively). Here, however, the regimes are determined by the curvature within the kink which is bounded by $I < 1/R_c^2$ (regime A), $1/R_v^2 > I > 1/R_c^2$ (regime B), and $I > 1/R_v^2$ (regime C), where I is given by eq. (40) and R_c , R_v are given by eqs. (9 - 10), respectively.

Explicitly, regime B is bounded by

$$\frac{2}{9} \left(\frac{R^2}{R_c^2} - 1\right) < (\xi^{\text{kink}})^2 < \frac{2}{9} \left(\frac{R^2}{R_v^2} - 1\right), \quad (47)$$

while regime A applies at the lower bound and regime C in the upper.

The bandgaps of the kink in the various regimes of bending can now be found by replacing $1/R^2$ in eqs. (12) with $(1 + \frac{9}{2}(\xi^{\text{kink}})^2)/R^2$ (eq. 41), which gives

$$E_g^{\text{kink}}(A) = \frac{|p|a\gamma}{R\sqrt{3}} + \frac{\gamma a^2}{16R^2} \text{sgn}[1-2p] \times \left(1 + \frac{9}{2}(\xi^{\text{kink}})^2\right) \cos 3\alpha, \quad (48)$$

where ξ^{kink} is given by eq. (38). In region *B*, eqs. (12) then give,

$$E_g^{\text{kink}}(B) = \frac{|p|a\gamma}{R\sqrt{3}} + \frac{(\text{sgn}[1-2p]\,\gamma a^2 - 16C_s)}{16R^2} \times \left(1 + \frac{9}{2}(\xi^{\text{kink}})^2\right)\cos 3\alpha, \quad (49)$$

and as before, in regime C,

$$E_g^{\text{kink}}(C) = 0. \tag{50}$$

Eqs. (48–50) give the bandgaps in the center of the kink for the respective radii ranges. Here, as in prebuckling regime, the bandgap as a function of bending can increase, decrease, or vanish, depending on the sign of p. The difference with the pre-buckling regime, however, is that here it is $\propto (\kappa - \kappa^{\rm cr})^{1/2}$; this behaviour is depicted for zigzag tubes where p = +1 (fig. 4a) and p = -1 (fig. 4b) and for metallic tubes of all chiral angles in fig. (3).

It may be useful to compare the bandgaps of the kink with the bulk (far from the kink – where $|z| \gg R$). Assuming that in the bulk ξ remains near criticality, the



Figure 4: Bandgaps and Fermi energies vs. bending for semiconducting zigzag tubes of the two types $(p = \pm 1)$. Following eq. (42), the initial gaps first decrease (left) / increase (right) $\propto \kappa^4$; at post-buckling it evolves faster: $\propto (\kappa - \kappa^{cr})^{1/2}$; at increased curvature – where the singlet band crosses the conduction band – the gap strongly downshifts in all cases (49); at this point, also the Fermi energy (C2) downshifts. The point of $E_F = 0$ in the plots corresponds to the value of E_F in graphene.

bandgap in regime B (eq. 49) compared with the bulk gives,

$$E_g^{\text{kink}} - E_g^{\text{bulk}} = -\frac{9C_s}{2R^2} \left((\xi^{\text{kink}})^2 - \left(\frac{2}{9}\right)^2 \right) \cos 3\alpha, \ (51)$$

where ξ^{kink} is given by eq. (38).

It is worth noting that for armchair tubes $(p = 0, \alpha = 30^{\circ})$, these equations (as well as the pre-buckling ones) predict no bandgaps. That holds, however, as long as the underlying assumption through this work – that the kink remains open [29] – holds.

IV. SUMMARY

We presented here a comprehensive theory of bandgaps in carbon nanotubes, including strong curvature and large non-homogeneous deformation. This theory reconciles the fundamental theory of bandgaps in nanotubes (appendix B) and its well-known corrections (at small deformation and small curvature), with contrasting results from a host of DFT computations of very small or highly deformed tubes; the present theory shows them to be special cases of the same general equations (12).

A formalism was derived to calculate the gaps due to a general non-homogeneous circumferential deformation (by starting with eq. 12 and making the replacement 33). We then applied this formalism to study bending, including buckling and a kink (with a caveat of staying within the transient regime, i.e: where the opposite walls of the kink do not touch).

The results detail the gap evolution under both weak and strong bending. In the pre-buckling regime, the bandgap shifts $\propto \pm \kappa^4$. A notable result is that, by the onset of buckling, the gaps of primary-metallic tubes, independent of chiral vector or radius, increase by a ratio of 2/9 compared with their un-bent value, (eq. 45 and fig. 3).

In the post-buckling regime ($\kappa > \kappa^{\rm cr}$, eq. A17), at first, the bandgap in the kink shifts $\propto \pm (\kappa - \kappa^{\rm cr})^{1/2}$, up-to the point where the singlet band crosses the conduction π band initiating a steep downshift to zero (eqs. 48-50, and fig. 4). The downshift in the bandgap is accompanied, in this regime, by a substantial downshift of the Fermi energy (appendix C).

V. APPENDICES

Appendix A: Brazier's Theory

This appendix has a number of relevant derivations from the elastic theory of thin cylindrical shells under pure bending. The theory was derived by Brazier [26]. For the exposition we follow reference [32].

Consider a slender cylindrical shell, initially straight, under bending. The strain profile is anti-symmetric about the neutral plane, compressive in the inner side and tensile in the outer; the energy per unit length is given by

$$U_{\rm z} = \frac{1}{2} I \kappa^2 \tag{A1}$$

where κ is the bending curvature and I is the second moment of area; for a perfectly circular cross-section

$$I \equiv I_0 = \pi R^3 t. \tag{A2}$$



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Figure 5: Bending causes the circular cross-section to become oval, turning $R \to R(1 + \xi \cos 2\theta)$ where ξ is the ovalization parameter; at $\xi = 2/9$ (blue shape) the tube reaches criticality and buckles.

where t is the thickness of the tube.

It will be shown below that by flattening the cross section the tube reduces its energy. It is reasonable to assume that small flattening can be expressed by an outof-plain displacement of the form

$$w = R\xi \cos 2\theta. \tag{A3}$$

where ξ (see fig. 5) is a dimensionless measure of the flattening, or ovalization. The curvature due to ovalization is

$$\delta\kappa_{\theta} = -\frac{\partial^2 w}{\partial y^2} - \frac{w}{R^2} = \frac{3\xi}{R}\cos 2\theta, \qquad (A4)$$

where $y = R\theta$ is the circumferential coordinate of the original circle, the second derivative is the usual curvature due to local changes of displacement while the second component is due to the change of radius. The inplain deformation v, together with w (A3), determines the circumferential strain $\epsilon_y = \partial v/\partial_y + w/R$. As a first approximation Brazier assumed the surface to be inextentional, ie: $\epsilon_y = 0$, yielding

$$v = -\frac{R}{2}\xi\sin 2\theta. \tag{A5}$$

where we assumed no net rotation by taking the constant of integration to be zero. It may be commented that for the electronic structure calculation we take a much looser requirement, instead of assuming the surface to be inextentional it is assumed only that the circumference is unchanged; i.e: $\oint \epsilon_y dy = 0$.

The second moment of area in eq. A1 is actually defined as

$$I = \int_0^{2\pi} Rt (R\sin\theta + \eta)^2 d\theta, \qquad (A6)$$

where t is the surface thickness and η is given by

$$\eta = w\sin\theta + v\cos\theta = -R\xi\sin^3\theta \qquad (A7)$$

where the last step used eqs. A3 and A5. Inserting this in eq. A6 we finally get

$$I = I_0 \left(1 - \frac{3}{2}\xi + \frac{5}{8}\xi^2 \right)$$
(A8)

where I_0 is given by A2. The strain energy per unit length (eq. A1) is then given by

$$U_{\rm z} = \frac{1}{2} \kappa^2 R^3 t E \pi \left(1 - \frac{3}{2} \xi + \left[\frac{5}{8} \xi^2 \right] \right)$$
(A9)

where E is the Young's modulus; the ξ^2 component in the square brackets is truncated in Brazier's analysis.

The energy per unit length due to Brazier deformation is

$$U_B = \frac{1}{2} D \int_0^{2\pi} \delta \kappa_\theta^2 R d\theta, \qquad (A10)$$

where $\delta \kappa_{\theta}$ is given by eq. A4 and D is the elastic rigidity given by

$$D = \frac{Et^3}{12(1-\nu^2)}.$$
 (A11)

Integration of eq. A10 gives

$$U_B = \frac{3\pi E t^3}{8(1-\nu^2)} \frac{\xi^2}{R}.$$
 (A12)

The total elastic energy per unit length is

$$U = U_{\rm z} + U_{\rm B}.\tag{A13}$$

Now ξ can be found by requiring $\partial U/\partial \xi = 0$. The result is

$$\xi = \frac{(1-\nu^2)R^4}{t^2}\kappa^2.$$
 (A14)

The values of ν and t were found by a number of groups (see table I), so we can write

$$\xi = BR^4 \kappa^2, \tag{A15}$$

where $B = 2.2(\text{Å}^{-2})$ by the values of [1] compared with $B = 1.57(\text{Å}^{-2})$ by [2].

The total energy (eq. A13) is finally given by

$$U = \frac{1}{2}R^{3}tE\pi\kappa^{2} - \left(\frac{3R^{7}(1-\nu^{2})E\pi}{8t}\right)\kappa^{4}$$
 (A16)

The tube begins to buckle when the bending moment $M = dU/d\kappa$, reaches maximum; hence the curvature at buckling point can be found by putting $d^2U/d\kappa^2 = 0$, which gives

$$\kappa^{\rm cr} = \frac{t}{3R^2} \sqrt{\frac{2}{1-\nu^2}}.$$
(A17)

 $\kappa^{\rm cr}$ is the critical curvature at the onset of buckling. Using the same values as for eq. A15, we find

$$\kappa^{\rm cr} = \frac{A}{R^2},\tag{A18}$$

where A = 0.316Å using the values of [1], compared with A = 0.376Å by [2]. MD simulations confirmed the general shape eq. (A18) and found A to range between 0.185 [3] and 0.387Å [1] Regardless of the actual numerical value of A, substituting A17 in A14 gives the exact ovalization parameter at buckling point

$$\xi^{\rm cr} = \frac{2}{9}.\tag{A19}$$

This is a remarkable result of Brazier's theory; it states that at the critical point of buckling, all tubes, independent of thickness, radius or Young's modulus, become ovalized by the same ratio of 2/9.

Appendix B: The fundamental bandgap

The band structure of carbon nanotubes is based on graphene's band structure [33] sliced-up with lateral quantization lines. The exact wrapping of graphene into a nanotube is determined by the tube's chiral integers (n, m) [34]. Now, any two integers can be related by other two integers (q, p), such that,

$$n - m = 3q + p \tag{B1}$$

where p takes one of the values (0, +1, -1).

The lateral k-vectors, k_y , must lie on quantization lines given by

$$k_y = \frac{\nu}{R}$$
, where $\nu = 0, \pm 1, \pm 2, \cdots$. (B2)

Graphene's Fermi surface is found by the k-vectors that solve,

$$\sum_{j=1}^{3} \gamma_j e^{i\vec{k}\cdot\vec{l}_j} = 0 \tag{B3}$$

where γ_j are the three overlap integrals of the nearest π -orbitals, and l_j are their bond vectors, given by

$$\vec{l}_{1} = \frac{a^{2}}{4\pi R} \left((n+m)\hat{y} - \frac{1}{\sqrt{3}}(n-m)\hat{z} \right),$$

$$\vec{l}_{2} = \frac{a^{2}}{4\pi R} \left(-m\hat{y} + \frac{1}{\sqrt{3}}(2n+m)\hat{z} \right),$$

$$\vec{l}_{3} = \frac{a^{2}}{4\pi R} \left(-n\hat{y} - \frac{1}{\sqrt{3}}(n+2m)\hat{z} \right),$$
 (B4)

where (n, m) are the chiral integers, and (\hat{y}, \hat{z}) are the circumferential and axial coordinates, respectively.

Now under the assumption that graphene is isotropic, we can take all

$$\gamma_j \equiv \gamma.$$
 (B5)

The solution of eq. (B3) is then given by the two points,

$$\vec{K}_{F1} = \frac{1}{3R} \left((m+2n)\hat{y} + m\sqrt{3}\hat{z} \right)$$
$$\vec{K}_{F2} = \frac{1}{3R} \left((n-m)\hat{y} + (m+n)\sqrt{3}\hat{z} \right)$$
(B6)

These are the Fermi points of graphene, in the nanotube's natural coordinates.

The spectrum near a Fermi point is linear,

$$E = \pm \frac{\sqrt{3}}{2} a\gamma \,\delta k,\tag{B7}$$

where E is the energy above the Fermi level and δk is the distance from the nearest Fermi point.

Substituting (B1) in (B6) reveals that when p = 0they do lie on a quantization line (B2) – these tubes are thus, according to zone-folding, metallic. For $p = \pm 1$, on the other hand, the distance to the nearest quantization line is 1/3R; these tube are thus semiconducting with an energy gap given by substituting 1/3R in the linear spectrum (B7),

$$E_g = |p| \frac{a\gamma}{\sqrt{3}R}.$$
 (B8)

Appendix C: Fermi energy and C_s

The lowering of the singlet (S) band below the conduction (π) band causes, as DFT simulations demonstrate (table IV), a downshift not only of the bandgap but also the Fermi energy. While it is generally understood to be a result of a large circumferential curvature of small tubes, we wish to quantify it by extrapolating the published data for tubes in this radii range (regime B: section II C).

In this regime, the bandgap depends on the position of the singlet band above the Fermi points (eq. 6), which depends, in turn, on the proportionality factor C_s . The DFT bandgaps of semiconducting, straight and undeformed tubes (table II) in this regime ($R \leq 4\text{\AA}$) give $C_s \approx 8(eV \cdot \text{\AA}^2)$.

ξ	11, 0	10, 0	8,0
0	0.82	0.87	0.487
0.0625	0.8	0.85	0.41
0.125	0.75	0.825	0.3
0.1825	0.65	0.7	0.075
0.25	0.525	0.55	0
0.3125	0.375	0.32	0
0.375	0.225	0.075	0
0.4375	0	0	0

Table III: Energy gaps (eV) in the respective tubes as a function of the deformation parameter ξ ; data extracted from figure 1a in ref. [31]



Figure 6: Bandgap vs. total circumferential curvature square for the zigzags (11,0), (10,0) and (8,0). Data points are given in table (III); lines are plotted by eq. (12) (regime B) where $1/R^2$ was replaced by I (see eq. 40) with $C_s \sim 8 - 10$ (eV·Å²). The crosses label the critical points of buckling (given by eq. A18).

Moving to Fermi energy, DFT simulations show (table IV) that the Fermi-energy of tubes in this regime are downshifted too in $\propto 1/R^2$ (fig. 7), while larger tubes have it identical to graphene. The S-band in this regime is the effective conduction band, and thus, the Fermi energy lies in the middle between it and the valence π -band (which is not shifted),

$$E_F = \frac{1}{2}(E_s + E_v) = \frac{1}{2}\left(E_s - \frac{E_g^{\pi}}{2}\right),$$
 (C1)

where the singlet band energy E_s is given by eq. (6) and the pure π bandgap E_g^{π} is given by eq. (4). This gives explicitly,

$$E_F = c_1 - |p| \frac{\gamma a}{4\sqrt{3}R} - \frac{\cos 3\alpha}{R^2} \left(c_2 + \operatorname{sgn}[1 - 2p] \frac{\gamma a^2}{64} \right). \quad (C2)$$

where the constants are

$$c_{1} = \frac{E_{g}(10,0)}{4} + \frac{C_{s}}{2R_{10,0}^{2}} \approx 0.53 \, eV$$
$$c_{2} = \frac{C_{s}}{2} \approx 4 \, eV \cdot \text{\AA}^{2},$$

where we used $E_g(10, 0) = 0.87 \, eV$ (table II), $R_{10,0} = 4$ Å, and $C_s = 8 \, eV \cdot Å^2$.

n,m	ref. [12]	ref. $\left[13\right]$
4, 0	-1.23	-1.29
5,0	-0.78	-0.64
7,0	-0.21	-0.38
8,0	-0.02	-0.14
10, 0	0.06	-0.04

Table IV: Fermi energies (in eV) of semiconductingzigzag tubes. What was actually computed are

work-functions, $(WF)^{\text{tube}}$; Fermi energies shown here were then found relative to graphene by $E_F^{\text{tube}} = (WF)^{\text{graphene}} - (WF)^{\text{tube}}$, where $(WF)^{\text{graphene}} = 4.55 \text{eV}$ in ref. [12] and 4.66 eV in ref. [13].

Now this can be further simplified if we neglect the second term in the parenthesis in eq. (C2) and, since $E_F(R \ge R_c) = 0$, following the downshift of the singlet band E_s (eq. 6),

$$E_F = \frac{C_s}{2} \left(\frac{1}{R_c^2} - \frac{1}{R^2} \right) \cos 3\alpha, \qquad R \le R_c, \qquad (C3)$$

where R_c is given, by eq. (9); for zigzag tubes ($\alpha = 0$), $R_c = R(10,0) = 4$ Å. Eq. (C3) is depicted in fig. (7) for semiconducting zigzag tubes in this range, together with DFT data.

Considering circumferentially ovalized tubes, following our procedure we first replace $1/R^2$ with $(1 + 9\xi^2/2)/R^2 \equiv I$ (eq. 40), where ξ is the ovalization parameter. Explicitly,

$$E_F(\xi) = \frac{C_s}{2R^2} \left(\left(\frac{R}{R_c}\right)^2 - 1 + \frac{9}{2}\xi^2 \right) \cos 3\alpha, \quad R \le R_c.$$
(C4)

Comparing this with the DFT bandgaps (table III), plotted in fig. (6), yields $C_s \sim 8$ (eV·Å²).

Finally, the fact that the analytic treatment in this work coincides with DFT on three different quantities: bandgaps of straight tubes, ovalized tubes, and Fermi energy of straight tubes, by having a single adjustable parameter, $C_s \sim 8 \text{ (eV-} \text{Å}^2)$, is, in our opinion, a strong indication of the correctness of this treatment.

nanotubes, The Journal of Chemical Physics **124**, 024709 (2006).

- [13] B. Shan and K. Cho, First principles study of work functions of single wall carbon nanotubes, Phys. Rev. Lett. 94, 236602 (2005).
- [14] G. Sun, J. Kurti, M. Kertesz, and R. H. Baughman, Variations of the geometries and band gaps of single-walled carbon nanotubes and the effect of charge injection, J. Phys. Chem. B 2003 107, 6924 (2003).
- [15] K. Kato and S. Saito, Geometries, electronic structures and energetics of small-diameter single-walled carbon nanotubes, Phycica E 43, 669 (2011).
- [16] V. Zólyomi and J. Kurti, First-principles calculations for the electronic band structures of small diameter singlewall carbon nanotubes, Phys. Rev. B 70 (2004).
- [17] M. O. et al., Energy gaps in metallic single-walled carbon nanotubes, Science **292(5517)**, 702–705 (2001).
- [18] X. B. e. a. Blase, Hybridization effects and metallicity in small radius carbon nanotubes, Phys. Rev. Lett. 72, 1878–1881 (1994).
- [19] A. Kleiner and S. Eggert, Curvature, hybridization, and stm images of carbon nanotubes, Phys. Rev. B 64, 113402 (2001).
- [20] D. H. Cao Jien, Wang Qian, Electromechanical properties of metallic, quasimetallic, and semiconducting carbon nanotubes under stretching, Phys. Rev. Lett. 90 (2003).
- [21] S. M. Umeno Yoshitaka, Sato Masanobu and S. Hiroyuki, Buckling induced band gap modulation in zigzag carbon nanotubes, Phys. Rev. B 100(15), 155116 (2019).
- [22] I. S., B. C., M. A., and B. J., Structural flexibility of carbon nanotubes., The Journal of Chemical Physics 104, 2089 (1996).
- [23] W. C. G. et. al., Buckling behavior of carbon nanotubes under bending: From ripple to kink, Carbon 102, 224 (2016).



Figure 7: Fermi energy of a number of small zigzag tubes vs. $1/R^2$. The line is a plot of eq. (C3) with $C_s = 8 (\text{eV} \cdot \text{Å}^2)$. Data is in table IV where red dots taken from ref. [12], green dots from [13].

- Y. B. I., B. C. J., and J. Bernholc, Nanomechanics of carbon tubes: Instabilities beyond linear response, Phys. Rev. Lett. **76** (1996).
- [2] T. Zhan, chun Ou, and Y. Z. Can, Single walled and multiwalled carbon nanotubes viewed as elastic tubes with the effective young's moduli dependent on layer number, Phys. Rev. B 65, 233407 (2002).
- [3] C. Guoxin and C. Xi, Buckling of single-walled carbon nanotubes upon bending: Molecular dynamics simulations and finite element method., Phys. Rev. B 73, 155435 (2006).
- [4] Y. Zhang, C. M. Wang, W. H. Duan, Y. Xiang, and Z. Zong, Assessment of continuum mechanics models in predicting buckling strains of single-walled carbon nanotubes, Nanotechnology **20** (2009).
- [5] R. A. J. et. al., Electronic and lattice properties of carbon nanotubes, J. Phys. Soc. Jpn. 63, 2252 (1994).
- [6] C. T. W. et al., Clusters and nanostructured materials, Edited by P. Jena and S. Behera (Nova, New York, 1996) , 231 (1996).
- [7] C. L. Kane and E. J. Mele, Size, shape, and low energy electronic structure of carbon nanotubes, Phys. Rev. Lett. 78, 1932 (1997).
- [8] L. Yang and J. Han, Electronic structure of deformed carbon nanotubes, Phys. Rev. Lett. 85, 154 (2000).
- [9] A. Kleiner and S. Eggert, Band gaps of primary metallic carbon nanotubes, Phys. Rev. B **63**, 73408 (2001).
- [10] A. Rochefort, D. R. Salahub, and P. Avouris, The effect of structural distortions on the electronic structure of carbon nanotubes., Chemical Physics Letters 297 (1998).
- [11] A. Rochefort, P. Avouris, F. Lesage, and D. Salahub, Electrical and mechanical properties of distorted carbon nanotubes., Phys. Rev. B. 60, 13824 (1999).
- [12] V. Barone, J. E. Peralta, J. Uddin, and G. E. Scuseria, Screened exchange hybrid density-functional study of the work function of pristine and doped single-walled carbon

- [24] G. Alessandra, G. Andrea, and S. Ginevra, Buckling and post-buckling analysis of single wall carbon nanotubes using molecular mechanics., Applied Mathematical Modelling (2020).
- [25] H. Shima, Buckling of carbon nanotubes: A state of the art review., (2018).
- [26] L. G. Brazier, On flexure of thin cylindrical shells and other thin sections, Proceedings of the royal society of London, series A 116, 104 (1927).
- [27] A. G. Mamalis, b E Manolakod, K. Baldoukas, and G. L. Viegelahn, Deformation characteristics of crashworthy thin-walled steel tubes subjected to bending, Proc Instn Mech Engr 203 (1989).
- [28] V. T. Z. L. C., Mechanism of bending with kinking of a single-walled carbon nanotube., Phys. Rev. B 69, 115410 (2004).
- [29] A. Kutana and K. P. Giapis, Transient deformation

regime in bending of single-walled carbon nanotubes., Phys. Rev. Lett. **97** (2006).

- [30] A. Barboza, A. Gomes, B. Archanjo, P. Araujo, A. Jorio, M. Ferlauto, A.and Mazzoni, H. Chacham, and B. Neves, Deformation induced semiconductor-metal transition in single wall carbon nanotubes probed by electric force microscopy, Phys. Rev. Lett. **100**, 2568049 (2008), flattening of the tube with AFM showed that it becomes semiconducting.
- [31] B. S. et al., First-principle study of band-gap change in deformed nanotubes, App. Phys. Lett. 87, 173109 (2005).
- [32] C. R. Calladine, *Theory of shell structures* (Cambridge University Press, 1989).
- [33] P. R. Wallace, The band theory of graphite, Phys. Rev. 71 (1947).
- [34] S. R., D. G., and D. M. S., Physical Properties of Carbon Nanotubes (Imperial college Press, 1999).