Machine Learning Prediction Models for Solid Electrolytes based on Lattice Dynamics Properties

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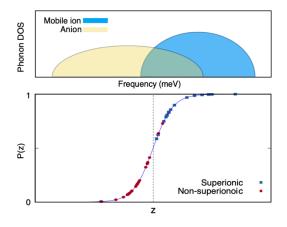
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ABSTRACT

Recently, machine-learning approaches have accelerated computational materials design and the search for advanced solid electrolytes. However, the predictors are currently limited to static structural parameters, which may not fully account for the dynamic nature of ionic transport. In this study, we meticulously curated features considering dynamic properties and developed machine-learning models to predict the ionic conductivity of solid electrolytes. We compiled 14 phonon-related descriptors from first-principles phonon calculations along with 16 descriptors related to structure and electronic properties. Our logistic regression classifiers exhibit an accuracy of 93 %, while the random forest regression model yields a root mean square error of 1.179 S/cm and R^2 of 0.710. Notably, phonon-related features are essential for estimating the ionic conductivity in both models. Furthermore, we applied our prediction model to screen 264 Licontaining materials and identified 11 promising candidates as potential superionic conductors.

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Concomitant to the sharp increase in global demand for electric vehicles, mobile electronics, and large energy storage, lithium-ion batteries (LIBs) have been intensively and extensively studied to improve their performance.^{1–3} However, organic liquid electrolytes, commonly used in LIBs due to their high ionic mobility, pose potential safety risks.^{4,5} Thermal instability of LIBs often arises from the breakage of the separator and electrochemical reactions within the electrolytes.

Inorganic solid-state electrolytes (SSE) have been investigated to mitigate the safety risks in the past decades.^{6–11} Compared to liquid electrolytes, SSEs are advantageous in electrochemical and thermal stability,¹⁰ as well as good cycle performance.^{9,11} It is also possible to operate under high voltage and achieve large energy density by use of the metallic lithium anode and the high voltage cathode.^{9,12} Examples include LISICON type (lithium superionic conductor), NASICON type (sodium superionic conductor), garnet, perovskites, and argyrodites materials such as Li₁₀GeP₂S₁₂,¹³ Li_{1.3}Al_{0.3}Ti_{1.7}(PO₄)₃,¹⁴ Li₇La₃Zr₂O₁₂,¹⁵ La_{0.5}Li_{0.5}TiO₃,¹⁶ and Li₆PS₅Br.¹⁷ However, SSEs exhibit relatively low ionic conductivity compared to organic liquid electrolytes, therefore many studies have searched for SSEs with high ionic conductivity and electrochemical stability.^{6,18,19}

Previous studies have suggested various parameters to control the ionic conductivity of solid electrolytes or superionic conductors. The static structure parameters⁶ such as bottleneck size,^{20,21} volume,^{22,23} and anion sublattice²³ have been reported to be related to ionic transport and activation energy. Not only static properties but also dynamic properties have been suggested as important features for ion mobility.^{6,24–31} For example, low energy optical (LEO) phonons induce a decrease in the activation barrier of ionic transport.²⁵ Soft lattice modes related to the octahedral rotation are connected to ionic diffusion in Ruddlesden-Popper phase Ln₂NiO₄₊₈ (Ln = La, Pr, Nd).²⁶

Phonon instability and migration profiles of Li₂O were discussed under both ambient and superionic phases.²⁷ Krauskopf *et al.* showed that the lattice softness reduces the activation energy for Na ion motion in Na₃PS_{4-x}Se_x.²⁸ The average phonon frequency of Li-ion was reported to correlate with the enthalpy of migration in LISICON,²⁹ and used as a descriptor for high-throughput screening.³²

Theoretical approaches to ionic conductivity have traditionally relied on direct methods, such as *ab initio* molecular dynamics (MD),^{33–35} despite being notably time-consuming. In recent years, the field of materials science has experienced a growing interest in the application of machine learning (ML) approaches,^{36–54} aimed at predicting properties based on the accumulated data. Consequently, several studies have applied various ML techniques to design potential SSEs.^{43–54} For example, Fujimura *et al.* performed the support vector regression to predict the ionic conductivity of LISICON.⁵⁴ Artificial neural network modeling was used to predict the Li diffusion barrier for LiMXO4.⁴⁷ Sendek *et al.* developed a classification model using various structural types of Li-ion conductors as a training set and proposed 21 promising candidates for SSE.⁴⁸ Unsupervised learning and compositionally restricted attention based networks (CrabNets) were also applied to distinguish fast Li-ion conductors.^{44,45}

In previous ML investigations,^{45–52} common features based on the atomic structures and chemical properties of components were used due to their straightforward quantification and ease of data collection. However, features related to lattice dynamics or phonon properties have recently shown importance in correlation with ionic diffusion,^{6,24–31} which have not been employed to predict the ion conductivity of the SSEs through ML approaches. Clarifying the mechanisms of ionic conductivities at fixed temperatures, such as room temperature, is challenging due to structural complexity and interactions. Therefore, it would be worth developing ML models with

features that include both dynamic and structural properties of materials and investigating their contribution to ionic conductivity.

In this study, we have developed ML models to predict the ionic conductivity of SSEs by utilizing the features related to not only static properties such as structural, electronic, and chemical features but also lattice dynamics. Figure 1 illustrates the schematic workflow of this work. First, we collected experimental ionic conductivity at room temperature (RT) and the corresponding crystal structures of Li and Na-based SSE materials from previous literature as shown in Table S1. For feature derivation, we performed density functional theory (DFT) calculations and obtained thirty features related to the phonon, electronic, and structural properties. The logistic regression (LR) classifier and random forest (RF) regression were employed for classification and regression models, respectively. The contribution of the features to estimate ionic conductivity is quantified by the Gini importance (GI) in the RF model. Furthermore, based on the developed LR model, we screened Li-SSE candidates containing O or S anions using the Materials Project (MP) database⁵⁵ and the phonon database at Kyoto University.^{56–58} Among them, 11 materials were identified as promising candidates for superionic conductors.

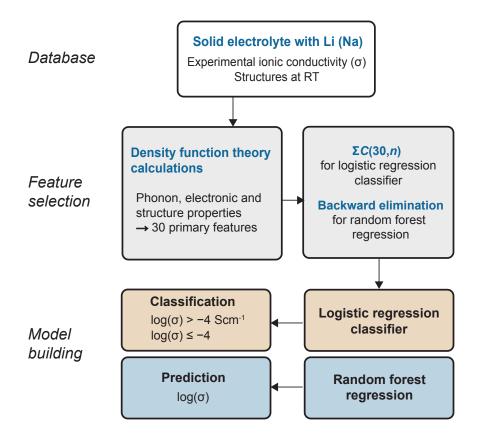


Figure 1. Schematic workflow of the ML model for ionic conductivity of Li and Na-based solidstate electrolytes.

To derive the features of electronic and structural properties, we conducted DFT calculations using the Vienna *ab initio* Simulation Package (VASP) code.^{59,60} All calculations are performed using Perdew-Burke-Ernzerhof (PBE) functional of generalized gradient approximation (GGA)⁶¹ and plane-wave energy cutoff of 650 eV. A *k*-point density of 5000/atom is used with respect to the corresponding system. The crystal structures were referenced from the MP⁵⁵ and Inorganic Crystal Structure Database (ICSD).⁶² The atomic and lattice structures are fully relaxed until the Hellmann-Feynman force is less than 0.001 eV/Å while maintaining the initial symmetry of the experimental structures. The disorder structures that have partial occupancies were regenerated in

ordered structure using Python Materials Genomics (pymatgen).⁶³ To obtain the phonon properties, we used the PHONOPY package, which applies a finite displacement approach and supercell calculations.⁶⁴ The force constants were calculated using the supercells with lattice parameters greater than 10 Å.

We employed LR and RF machine learning algorithms to classify and predict ionic conductivity. All models were developed and implemented through the application of the scikit-learn python library. We used the leave-one-out cross-validation (LOOCV) and evaluated models to suppress overfitting. The metric of accuracy and F1 score, and metrics of the root mean square error (RMSE) and coefficient of determination (R^2) were utilized for LR classification and RF regression models, respectively (See the Supporting information for the detailed information on accuracy and F1 score).

The LR model accepts any real value and generates a probability value between 0 and 1 through the logistic function $P(z) = \frac{1}{1+e^{-z}}$. This model is applied in predicting ionic conductivity using the equation, so-called z function, $z = w_0 + \sum_i w_i x_i$, where w_i and x_i denote the regression coefficient and the value of the feature, respectively. To find the best combination of features, we searched all combinations of 30 features, resulting in a total of $\Sigma {30 \choose n} = 1,073,741,824$ models.

We implemented the RF model,⁶⁵ an ensemble of models using decision trees. By constructing a forest comprising numerous decision trees, the RF model generates predictions by averaging the outcomes of each tree. The RF is preferred due to its ability to learn intricate non-linear dependencies and tolerance to data heterogeneity. The significance of features with respect to the output can be determined using GI. To obtain the optimal feature set while managing computational load, the backward feature elimination method was utilized. The process began with a model using the complete set of n features. Subsequently, all n-1 combinations of features were systematically tested, and the next set was constructed in each iteration by eliminating the least significant feature. The model performance at each cycle was determined based on the RMSE of the test set. The process terminates when the RMSE reaches its minimum, just before exhibiting an increase rate of 10 % in the next iteration.

Table 1. List of 30 features used in this work. The subscripts (superscripts) of M, A, C, Tot, and H denote the mobile ions (Li, Na), anion, cation, total atoms, and hybridization.

Features	Description
<ω> _M , <ω> _A , <ω> _{Tot}	Phonon band center of mobile ion, anion, and total atoms
$f_{ m max}^{ m M}, f_{ m max}^{ m A}, f_{ m max}^{ m H}$	Frequency at maximum phonon DOS of mobile ion, anion, and hybridization of mobile ion and anion below 30 meV
$P_{\max}^{M}, P_{\max}^{A}, P_{\max}^{H}$	Maximum phonon DOS below 30 meV per corresponding atoms
f_{img}	The existence of soft mode
R _{M/Tot} , R _{A/Tot} , R _{H/Tot}	The ratio of atom-projected phonon DOS per total phonon DOS below 30 meV
S	Vibrational entropy at RT
E _G	Energy band gap
$\Delta H_{ m f}$	Heat of formation
$V_{f.u},V_M$, V_A,V_{Tot}	Volume per formula units, mobile ion, anion, and total atoms
$d_{ extsf{M-A}}$, $d_{ extsf{M-M}}$	The minimum distance between Li (Na) and anion, Li (Na) and Li (Na)
N _{f.u}	Number of formula units in the conventional cell
ho	Density
т	Total mass
N _M	Number of mobile ions in formula unit
M-disorder, A-disorder,	Site atomic disorder of mobile ion, anion, cation, and total
C-disorder, Tot-disorder	atoms

Firstly, we compiled a list of 45 Li and Na-ion conductors along with their structures and ionic conductivity at RT, denoted as σ , as detailed in Table S1. Figure S1 shows the histogram of the dataset categorized by log(σ). Feature selection is one of the most important parts to improve a model performance. To consider multifaceted perspectives, we selected features incorporating phonon, structural, and electronic properties based on previous literatures.^{27,29,66–70} and physical

intuition. We also defined several features to quantitatively describe specific properties related to phonon DOS and disorders. Table 1 summarizes the selected 30 features.

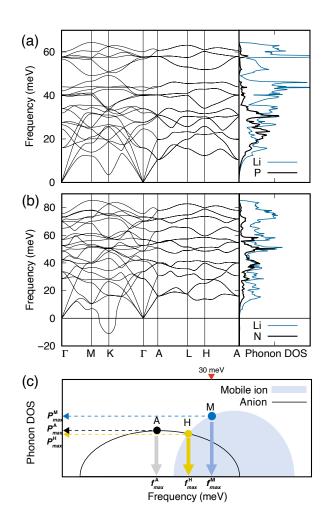


Figure 2. Phonon dispersions and density of states (DOS) of (a) Li₃P and (b) β -Li₃N as representatives. The negative phonon frequency indicates the phonon soft mode. (c) Schematic phonon DOS. $P_{\text{max}}^{\text{M}}$, $P_{\text{max}}^{\text{A}}$, and $P_{\text{max}}^{\text{H}}$ indicate the maximum phonon DOS below 30 meV for mobile ion (Li or Na), anion, and their intersection (hybridization) region. $f_{\text{max}}^{\text{M}}$, $f_{\text{max}}^{\text{A}}$, and $f_{\text{max}}^{\text{H}}$ are the corresponding frequencies to $P_{\text{max}}^{\text{M}}$, $P_{\text{max}}^{\text{A}}$, and $P_{\text{max}}^{\text{H}}$, respectively.

To consider features related to lattice dynamics, we analyze phonon bands and include various representations of phonon characteristics. Figure 2(a) and (b) show the phonon dispersion curves of Li₃P ($\sigma = 1.0 \times 10^{-3}$ S/cm) and β -Li₃N ($\sigma = 2.085 \times 10^{-4}$ S/cm), respectively, as representative examples. First, we used < ω > to denote the average vibrational frequency so-called 'phonon band center' following the previous study.²⁹ The quantity of < ω > is expressed as $<\omega$ > = $\frac{\int \omega \times DOS(\omega)d\omega}{\int DOS(\omega)d\omega}$. < $<\omega$ >_M, $<\omega$ >_A, and $<\omega$ >_{Tot} correspond to the phonon band centers of mobile ion (Li or Na), anion, and total atoms, respectively.

In previous studies, the glassy Li₃P₄ phase has been observed to exhibit a lower peak of Li and larger overlap with anion spectra than crystalline γ -Li₃P₄.⁶⁶ These characteristics were thought to enhance ion mobility at low temperatures by promoting the paddlewheel dynamics. However, other mechanisms to enhance ion mobility related to such phonon spectra feature have been proposed recently,^{31,71} such as anharmonic phonon coupling. We constructed features that relate to the phonon peak position and defined a feature associated with the overlapping phonon DOS of Li (Na) and anion. Specifically, the features, f_{max}^{M} and f_{max}^{A} correspond to the frequency at the maximum phonon DOS of mobile ion and anion, respectively. The frequency at the phonon peak for 'hybridization', *i.e.*, the overlap region of mobile ion and anion, is denoted as f_{max}^{H} , as shown in Figure 2(c). P_{max}^{M} , P_{max}^{A} , and P_{max}^{H} are defined as phonon DOS values at f_{max}^{M} , f_{max}^{A} , and f_{max}^{H} per corresponding atoms.

In addition, we restricted the highest phonon frequency to 30 meV, which is close to the RT energy scale of ~25 meV. Phonons with energies higher than the RT energy of 25 meV have significantly smaller average number of phonons at RT because the average number of phonons,

 $\langle n \rangle$ at fixed temperature follows the Planck distribution, $\langle n \rangle = \frac{1}{\exp(\frac{\hbar\omega}{k_BT} - 1)}$. This restriction also

emphasizes the low frequency region that is related to lattice softness.

We further selected features that can be associated with lattice softness. The feature f_{img} denotes the existence of soft mode in the phonon bands since it can be considered a precursor to ionic diffusion.²⁷ We designated a value of 0 if there are no imaginary frequencies. Values 1 and 2 were assigned if soft modes originate from mobile and other ions, respectively. Furthermore, we considered additional phonon-related properties such as the phonon DOS of mobile ion, anion, and hybridization region per total phonon DOS ($R_{M/Tot}$, $R_{A/Tot}$, $R_{H/Tot}$) below 30 meV, along with vibrational entropy, S at RT.⁷⁰ The ratio of the phonon DOS below 30 meV indicates how the phonon DOS is relatively distributed in the low frequency region. The softer lattice tends to exhibit higher vibrational entropy.⁷⁰

For electronic and structural properties, we selected features, similar to previous literature,^{47,48} such as band gap (E_G), the heat of formation (ΔH_f), volume per formula units, mobile ion, anion, and total atoms ($V_{f.u}$, V_M , V_A , V_{Tot}), the minimum distance between Li (Na) and anion (d_{M-A}), Li (Na) and Li (Na) (d_{M-M}), and the number of formula units ($N_{f.u}$) in the conventional cell. In addition, we included the properties derived from the chemical composition of materials such as density (ρ),⁴⁶ mass (m), and the number of mobile ions in the formula unit cell (N_M).

We also incorporated site-disorder properties into the feature set, as activation energies for Liion diffusion are expected to decrease with increasing site-disorder.^{67–69} We quantified disorders using site occupancy and the Wyckoff site number. M-disorder, A-disorder, C-disorder, and Totdisorder denote the amount of the mobile ion, anion, cation, and total site-disorder in the given structure, respectively. Computational details for quantifying the disorder in an example system are described in the Supporting information.

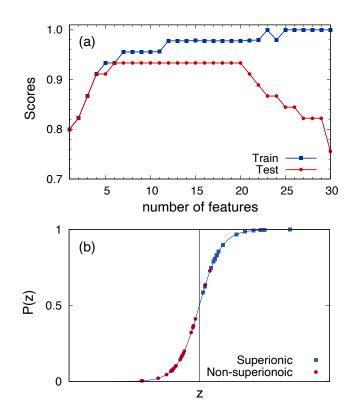


Figure 3. Prediction performance of the LR models. (a) The highest accuracy scores according to the number of features. (b) The classification performance using 6 features with an accuracy of 0.933 and an F1 score of 0.927. The blue squares and red dots represent the superionic and non-superionic materials based on experimental observations, respectively.

In the LR approach for the binary classification model, materials were categorized as superionic and non-superionic based on a threshold value of 10^{-4} S/cm, and classified into binary outputs of 1 (log (σ) > -4 Scm⁻¹) and 0 (log (σ) \leq -4 Scm⁻¹) as suggested by the previous study.⁴⁸ To search for the best combination of predictors, we evaluated the prediction performance by considering all possible combinations of the 30 primary features. Figure 3(a) shows the highest scores of our models depending on the number of features. The score of the test set increases sharply up to a score of 0.933 and remains constant until 20 features. The performance of the test set deteriorates with more than 20 features, unlike the training set displaying high accuracy, indicating overfitting. Using a single feature set, the best model achieves an accuracy of 0.8 with the $N_{f.u}$ feature. Combinations with other features further enhance performance, reaching up to 0.933 with 6 features.

Table 2 shows the best predictors with 2 to 6 combinatorial features and their accuracies. Even with just four features, the prediction performance presents an accuracy greater than 0.9, as depicted in Figure 3(a). The best combinations of 4 features, scoring an accuracy of 0.911, include $N_{f.u}$, $R_{M/Tot}$, C-disorder, and N_M ($V_{f.u}$). $N_{f.u}$, $R_{M/Tot}$, and C-disorder consistently appear in the best combinations ranging from 4 to 19 features. The combination of $R_{H/Tot}$, S, and A-disorder, together with $N_{f.u}$, $R_{M/Tot}$, and C-disorder exhibit the highest performance with an accuracy of 0.933. Conversely, combinations of 4 and 6 features excluding phonon-related attributes result in accuracies of 0.867 and 0.889, respectively. These results suggest the influential role of phonon characteristics in enhancing the performance of the ML model.

Table 2. The optimal	feature	set and	its corres	ponding	test accuracy

Number of features	Best Combinations	Accuracy
2	N _{f.u} , C-disorder N _{f.u} , A-disorder N _{f.u} , Tot-disorder	0.822
3	$N_{f.u}$, C-disorder, E_G $N_{f.u}$, C-disorder, N_M $N_{f.u}$, $R_{M/Tot}$, f_{img} , $N_{f.u}$, f_{max}^A , S	0.867
4	$N_{f.u}, R_{M/Tot}, C$ -disorder, N_M $N_{f.u}, R_{M/Tot}, C$ -disorder, $V_{f.u}$	0.911
5	$N_{f.u}$, $R_{M/Tot}$, C-disorder, A-disorder, S $N_{f.u}$, $R_{M/Tot}$, C-disorder, d_{M-M} , m	0.911
6	Nf.u, R _{M/Tot} , R _{H/Tot} , C-disorder, A-disorder, S	0.933

with respect to the number of features.

Figure 3(b) illustrates the logistic function P(z) obtained using the 6 features of N_{f.u}, R_{M/Tot}, R_{H/Tot}, C-disorder, A-disorder, and S. A value of P(z) greater than (less than) 0.5 indicates the prediction of superionic (non-superionic) behavior. Out of the 45 materials, three are misclassified: Li₄BN₃H₁₀, Li₆PS₅I, and Li₂CaN₂H₂. The confusion matrix is shown in Figure S2.

$$z = -0.427 \text{ N}_{f.u} + 4.523 \text{ R}_{M/Tot} + 2.561 \text{ R}_{H/tot} + 1.305 \text{ C} - \text{disorder} + 1.305 \text{ C} - 1$$

$$2.220 \text{ A-disorder} + 0.007 \text{ S} - 2.552 \qquad (1)$$

Equation (1) presents the *z* function for the best combinations of 6 features in Table 2. By examining the signs of the coefficients w_i in the *z* function, we can infer their influence on the likelihood of superionic behavior. From a structural perspective, crystal structures with fewer number of formula units tend to exhibit high ionic conductivity ($w_{N_{fu}} < 0$). The presence of cation and anion disorders positively influences fast ionic transport ($w_{C-disorder} > 0$, $w_{A-disorder} > 0$). Regarding dynamic properties, softened lattice vibrations of mobile ions (Li or Na) contribute to the increased probability of superionic conductivity ($w_{R_{M/Tot}} > 0$). A larger overlap of mobile ion and anion phonon DOS promotes superionic conductor ($w_{R_{H/Tot}} > 0$). Furthermore, higher vibrational entropy that is associated with the softness of the lattice enhances the fast ionic conductivity ($w_S > 0$). The Pearson correlation coefficients of N_{fu}, $R_{M/Tot}$, $R_{H/Tot}$, C-disorder, A-disorder, and S to the ionic conductivity are -0.400, 0.357, 0.296, 0.349, 0.238, and 0.493, respectively, indicating consistent positive/negative relationships with the LR results (See Table S3).

Table 3 presents the GI derived from the RF algorithm, which evaluates the significance of each feature on the output. Among these six features, $R_{H/Tot}$, $R_{M/Tot}$, and $N_{f.u}$ are the most important features for classifying superionic conductors, each with similar GIs of > 0.2. Following closely is vibrational entropy with a GI of 0.146. Conversely, disorder-related features contribute minimally

to the classification of ionic conductivity. It is noteworthy that features related to phonons exhibit high GI values, showing their relevance to fast ion transport.

Rank	Classificat	Classification		Regression	
1	R _{H/Tot}	0.236	f_{\max}^{A}	0.312	
2	R _{M/Tot}	0.235	P_{\max}^{M}	0.233	
3	$N_{f.u}$	0.227	$N_{f.u}$	0.216	
4	S	0.146	N_M	0.174	
5	C-disorder	0.092	C-disorder	0.065	
6	A-disorder	0.064			

Table 3. The Gini importance of each feature from the best LR and RF model.

We further developed the ensemble regression model of RF to predict the ionic conductivity and assess the contribution of each feature to estimating it. Figure 4(a) shows the prediction performance with RMSE and R^2 values according to the number of features. Each point represents the RMSE and R^2 obtained from the best combination of features using the backward feature elimination method. Among models with a single feature, the best performance is achieved when using f_{max}^A , with an RMSE of 1.696 S/cm and R^2 of 0.296, respectively. As the number of features increases, the RMSE values decrease to 1.179 S/cm and then increase with more than 5 features, reaching up to 1.645 S/cm with 30 features. The combination of five features, f_{max}^A , P_{max}^M , $N_{\text{f.u}}$, N_M , and C-disorder, demonstrates the best prediction performance an RMSE of 1.179 S/cm and R^2 of 0.710, as shown in Figure 4(b). The best performance of the RF models is not better than the classification model using LR despite a high R^2 of 0.96 in the training set. This would be because accurately predicting ionic conductivity is more complex than classifying properties.

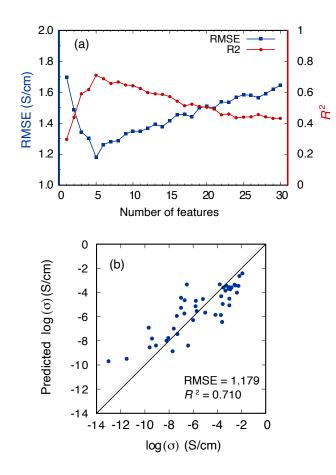


Figure 4. Prediction performance of the RF. (a) RMSE and R² depending on the number of features.
(b) Comparison of the predicted and experimental ionic conductivity log (*σ*).

Table 3 represents the GIs of f_{max}^{A} , P_{max}^{M} , $N_{\text{f.u}}$, N_{M} , and C-disorder. The frequency at maximum phonon DOS of anion below 30 meV, f_{max}^{A} , is the highest-ranked feature with a GI of 0.312. The P_{max}^{M} and $N_{\text{f.u}}$ rank second and third, contributing comparably to this model, with similar GIs of 0.233 and 0.216, respectively. Mobile ion density (N_{M}) is the fourth important feature, while C-disorder is the least important in predicting ionic conductivity.

Note that two phonon-related features (f_{max}^{A} , P_{max}^{M}) are included as important features in the best performance regression model. Like the classification model, the most significant properties in the regression model are associated with the phonon properties according to the GIs. Considering the

dynamic attributes of both mobile ions and anions improves the performance of identifying materials with high ionic conductivity. These results are consistent with the previous literature reporting the correlation with lattice dynamics on ionic transport,^{6,24–31} and our results further show their statistical significance through ML analysis. N_{f.u} and C-disorder features are also included in the best feature combination as in the LR model, indicating their role in determining ionic conductivity. As shown in the GI and *z* function, the positive influence of disorder-related features is consistent with previous studies.^{67–69}

MP ID	Composition	Space group
4527	Li_8SnO_6	R-3
6844	$KLi_3Si_{12}Sn_2O_{30}$	P6/mcc
8610	Li ₈ PtO ₆	R-3
14364	Cs ₂ LiAsO ₄	$Cmc2_1$
16055	$KLi_3Zr_2Si_{12}O_{30}$	P6/mcc
17208	$K_2Li_{14}Zr_3O_{14}$	Immm
18711	LiNdP ₄ O ₁₂	C2/c
504806	$K_2Li_{14}Pb_3O_{14}$	Immm
504810	$Rb_2Li_{14}Pb_3O_{14}$	Immm
774749	RbNa ₃ Li ₁₂ Ti ₄ O ₁₆	I4/m
753546	Li_8TiS_6	P63cm

Table 4. Candidates for superionic conductors screened by the LR model.

Finally, we apply our LR model to search for new SSE candidates. All structures and phonon data are acquired from the MP⁵⁵ and phonon database at Kyoto University.^{56–58} We have screened all the Li-based materials from the phonon database with the S or O anions. The materials that have a small gap (< 1.0 eV) are eliminated to minimize the electronic conductions as electrolytes. We used the materials that satisfy the convex hull energy (E_{hull}) is 0 eV/atom to prevent sample

decomposition. From these requirements and restrictions, we compiled 259 Li-contained materials and added additional 5 materials (LiErSe₂, LiHo₃Ge₂O₈F₂, LiBiF₄, LiCaGaF₆, LiInF₄) from the previous MD report⁴⁹ for comparison.

By screening 264 datasets based on our LR model with six features ($N_{f.u}$, $R_{M/Tot}$, $R_{H/Tot}$, C-disorder, A-disorder, S), we identified 11 materials out of 264 (~4%) that may exhibit superionic characters as listed in Table 4. Among the predicted candidates, Li_8SnO_6 has been explored as an electrode material for LIBs due to its high lithium content,^{72–74} and DFT studies reported the migration energy ranging from 0.20 eV to 1.06 eV across different paths and methods.^{72,73} Other candidates listed in Table 4 have not yet been identified through either DFT-MD or experiments to the best of our knowledge. To further validate the performance of our model, we compared the results to the previous report using the DFT-MD simulations.⁴⁹ Our prediction matches the calculated ionic conductivity with 87.0 % accuracy as shown in Table S4. Although we observed a few false negative results compared to the previous MD results, we anticipate that our screening model with phonon features has prediction power and will facilitate the discovery of superionic conductors.

In conclusion, we have developed ML models utilizing LR and RF algorithms to predict the ionic conductivity of SSE materials. Our approach involved careful selection and quantification of features from both dynamic and static properties using the DFT calculations and experimental structures. The LR model exhibits an accuracy of 93 % with six features: $N_{f.u}$, $R_{M/Tot}$, $R_{H/Tot}$, C-disorder, A-disorder, and S. A structure with a small number of formula units, lattice softness around Li (Na) ions, a large overlap of phonon DOS in mobile ions and anions, high disorder of cations and anions, and high vibrational entropy promote superionic behavior. The RF model presents the prediction performance with an RMSE of 1.179 S/m and an R^2 of 0.710. Notably,

phonon-related features such as f_{max}^{A} and P_{max}^{M} , along with structure-related features including N_{f.u}, N_M, and C-disorder, play crucial roles in predicting ionic conductivity in the RF regression model. In both classification and regression models, phonon-related features exhibit higher GI, quantitatively indicating their strong relevance to ionic conductivity. Thus, our ML models demonstrate the significance of dynamic properties in predicting and understanding the ionic conductivity of materials. Finally, based on the developed models, we screened the 264 Li-based materials for SSE materials and suggested 11 candidates for superionic conductors. We believe that our understanding can guide the future design of advanced superionic conductors.

SUPPORTING INFORMATION

List of the training set, which includes composition, ionic conductivity at RT, and space group of materials, along with histograms of the dataset, an example for the disorder features, the confusion matrix, the Pearson correlation map of the features and conductivity, and ML screened results for comparison to previous MD study.

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